

**P-type thermoelectric properties of half-Heusler alloys  $\text{TiNi}_{1-x-y}\text{Co}_y\text{Sn}$   
( $0 \leq x \leq 0.1$ ,  $0 \leq y \leq 0.05$ ) below 800 K**

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Half-Heusler (hH) alloy  $\text{TiNiSn}$  is a thermoelectric material with high environmental compatibility, low cost, good mechanical properties, and chemical stability at high temperatures. There are many studies on the N-type thermoelectric properties of  $\text{TiNiSn}$ -based material, but few experimental reports on the P-type. Our research group reported the thermoelectric properties of  $\text{TiNi}_{1-x-y}\text{Co}_y\text{Sn}$  ( $x=0$ ,  $0 \leq y \leq 0.15$ ) [1]. However, the  $ZT$  of P-type is still lower than that of N-type  $\text{TiNiSn}$ -based materials, and the high electrical resistivity is found to be one of the reasons. In addition, it was found that Ni formed intrinsic defects in Co-substituted  $\text{TiNiSn}$  fabricated with stoichiometric composition. This defect acts as a donor impurity, emitting free electrons. Therefore, the present work aims to optimize the Ni composition ratio ( $x \neq 0$ ) in order to reduce defects, leading to an increase in hole density. This is expected to decrease electrical resistivity and improve the  $ZT$  of Co-substituted  $\text{TiNiSn}$ .

The raw materials were weighed to obtain 15 g of  $\text{TiNi}_{1-x-y}\text{Co}_y\text{Sn}$  ( $0 \leq x \leq 0.1$ ,  $0 \leq y \leq 0.05$ ) and melted under an Ar atmosphere. The samples were cut to measurable scale by wire Electrical Discharge Machining, annealing in vacuum-sealed quartz tubes for homogenization, and heat treatment. All the samples were subjected to powder X-ray diffraction measurements, and the crystal phases were identified by Rietveld analysis using the obtained diffraction patterns. The sample composition was confirmed to be almost nominal compositions by using ICP-AES. ResiTest8300 (TOYO Corporation) was used for electrical resistivity and Seebeck coefficient measurements from 80 K to 395 K, and a home-made apparatus was used for measurements from 395 K to 800 K. Thermal conductivity measurements from 300 K to 800 K were used by PEM-2 (ADVANCE RIKO, Inc.), and  $ZT$  up to 800 K was evaluated.

From crystal structure analysis, all samples are mostly composed of the hH phase. In addition, the fraction of the impurity phase increases with increasing  $x$ . Above  $x = 0.08$ , the impurity phase increased to around 5%, suggesting a deterioration of thermoelectric properties. From measurements of thermoelectric properties, electrical resistivity decreases with increasing  $x$ . Interestingly, the absolute value of the Seebeck coefficient remains unchanged up to  $x = 0.05$ . Therefore, at 800 K, the power factor increases from  $0.85 \text{ mWm}^{-1}\text{K}^{-2}$  to  $1.7 \text{ mWm}^{-1}\text{K}^{-2}$  as  $x$  increases from 0 to 0.03 for  $y = 0.05$ . The thermal conductivity increases with increasing  $x$ , which can be attributed to the disappearance of the phonon scattering and the decrease in electrical resistivity. As for  $ZT$ ,  $x = 0.03$ ,  $y = 0.05$  shows the largest  $ZT$  of 0.18 at 700 K in this study. This value is higher than a similar P-type hH alloy, e.g.,  $\text{ZrNi}_{0.98}\text{Co}_{0.02}\text{Sn}$  ( $ZT \approx 0.1$ ) at 700 K [2]. This result indicates that the decrease in the Ni composition ratio is effective in increasing  $ZT$  for P-type  $\text{TiNiSn}$ -based materials.

## References

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- [2] Xie, H. H., Yu, C., He, B., Zhu, T. J., Zhao, X. B., *J. Electron. Mater.* **2012**, 41, 1826-1830.